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## Effect of annealing on the Structure of thermal evaporated $\text{In}_2\text{S}_3$ thin films

**Abstract.** Structural studies on  $\text{In}_2\text{S}_3$  thin films deposited by vacuum thermal evaporation on glass substrates at a temperature of 240°C, followed by annealing at 330°C and 400°C are presented. It was shown that the films were of amorphous in nature before annealing and after annealing the films became polycrystalline and showed  $\beta$ - $\text{In}_2\text{S}_3$  structure. The grain size increased and followed the usual crystal growth process as the annealing temperature increased. The annealing-induced changes of surface roughness were characterized by atomic force microscopy.

**Streszczenie.** Przedstawiono wyniki badania struktury cienkich warstw  $\text{In}_2\text{S}_3$  naniesionych metodą termicznego osadzenia na szkło w próżni w temperaturach od 240°C i kolejnym wygrzewaniem w temperaturach 330°C i 400°C. Ustalono że warstwy miały amorficzną naturę przed wygrzewaniem, po wygrzewaniu wykazywały strukturę polikrystaliczną typu  $\beta$ - $\text{In}_2\text{S}_3$ . Rozmiar ziaren powiększał się na skutek aktywnej nuklearyzacji i to przyczyniało się do zwykłej krystalizacji ze wzrostem temperatury. Zmiany zostały przeanalizowane za pomocą AFM. (Wpływ obróbki temperaturowej na termicznie naparowane cienkie warstwy  $\text{In}_2\text{S}_3$ ).

**Słowa kluczowe:**  $\text{In}_2\text{S}_3$ , cienkie warstwy, XRD widma, Raman widma, AFM obraz powierzchni

### Introduction

Thin films of III-VI materials have found applications in many areas of science and technology that include optoelectronics and photovoltaics [1, 2]. Among them,  $\text{In}_2\text{S}_3$  films are excellent candidates for different applications due to their stability, attractive optical properties and photoconductor behavior [3-6]. Especially  $\text{In}_2\text{S}_3$  thin films are suitable for application in solar energy photovoltaic conversion [7-9]. It may be perspective to form buffer layers for  $\text{Cu}(\text{In,Ga})\text{Se}_2$  thin film solar cells, which may achieve efficiency up to 16.4% [8].  $\text{In}_2\text{S}_3$  is a n-type semiconductor that can exist in several polymorphs and this makes the physical properties of  $\text{In}_2\text{S}_3$  films significantly dependent on the preparation conditions. The future improvement of solar cells therefore requires the scrupulous characterization of thin film surface and micro-structural properties (including phase composition, grain size, and roughness).

X-ray diffraction (XRD) and Raman spectroscopy are well known as most powerful techniques for structural analysis of thin solid films. Surface roughness and topography, which are used to characterize contact surfaces, were examined via Atomic Force Microscopy (AFM) [10]. In practice, the most commonly used parameters for surface roughness description are average surface roughness  $R_a$  and root mean square roughness  $R_{\text{rms}}$  [11, 12]. However, there are also other roughness parameters that give better surface description like skewness of the roughness profile,  $R_{\text{sk}}$  and kurtosis of the roughness profile,  $R_{\text{ku}}$ . The reported data on the influence of annealing temperature on these roughness parameters of thermally evaporated  $\text{In}_2\text{S}_3$  films is very meager.

Earlier, the effect of the annealing temperature and the thickness of the thermally evaporated  $\text{In}_2\text{S}_3$  films on their optical properties was studied in [13, 14]. However, there is to date a lack of reported studies regarding influence of annealing temperature on the topography and surface roughness of thermally evaporated  $\text{In}_2\text{S}_3$  films. Hence, the present study is aimed to investigate the microstructural properties of  $\text{In}_2\text{S}_3$  films in terms of phase composition and surface topographical parameters.

### Method and calculation details

The samples of investigated films were produced by a vacuum thermal evaporation of polycrystalline  $\text{In}_2\text{S}_3$  powder onto glass substrates in the vacuum chamber (less than

$5 \cdot 10^{-4}$  Pa) at 240°C temperature and deposition rate of  $\sim 0.5$  nm/s. The thickness of the films used in this investigation was 1.2  $\mu\text{m}$ . The deposited films were subjected to annealing for 60 min at different temperatures, 330°C and 400°C. More detailed description of the synthesis conditions can be found in the papers [13, 14].

In the present investigation, a Siemens D-5000 X-ray diffractometer was used to investigate the crystal structure, present phases and other structural parameters. It uses  $\text{CuK}\alpha$  radiation ( $\lambda = 0.15405$  nm) with a nickel filter and the data was recorded over the scattering angle range, 10 – 60 degree. Raman spectra were obtained at a room temperature using a DILOR XY 800 spectrometer. For the excitation of Raman radiation, an Ar-laser with a wavelength of 514.5 nm was used. JEOL JSPM-4210 atomic force microscope was used to study the surface topographical parameters in the noncontact mode.

### Results and discussion

Fig. 1 shows XRD spectra of  $\text{In}_2\text{S}_3$  thin films deposited on glass substrates and annealed at 330°C and 400°C. It is seen that the films have an amorphous structure after the deposition. The absence of any of  $\text{In}_2\text{S}_3$  peaks in the as-deposited films is due to the low thermal energy, which is insufficient for significant crystallization [15, 16]. X-ray diffraction spectrum of the  $\text{In}_2\text{S}_3$  film annealed at 330°C showed low-intensity peaks related to the (214) and (324) crystalline planes of the tetragonal  $\text{In}_2\text{S}_3$  phase [17]. X-ray diffraction spectrum of the  $\text{In}_2\text{S}_3$  annealed at 400°C showed the appearance of high-intensity peak near  $2\theta = 33$  deg. corresponding to (220) crystalline plane of tetragonal phase [18]. Additionally, the XRD spectra of the films annealed at 400°C showed diffraction peaks that correspond to the (008), (115), (204), (311), (215) and (422) planes of the cubic  $\text{In}_2\text{S}_3$  [16]. Previously, co-existence of the cubic and tetragonal phases in  $\text{In}_2\text{S}_3$  layers grown by the chemical bath deposition was reported in [19]. Besides, a phase change in the evaporated  $\text{In}_2\text{S}_3$  films with the change of substrate temperature was also reported in [20].

The fact that the full width at half maximum (FWHM) decreased with the raise of annealing temperature is an indicator of improved crystalline quality of thin film. As seen from the Fig. 1, there was an intensity increase for XRD peaks with increasing annealing temperature. This may be due to thermal energy increase enough for recrystallization

and the grain growth. The crystallite sizes were calculated by the Scherrer formula from the peak FWHM [21]

$$D = \frac{0.94\lambda}{\beta \cos \theta},$$

where  $D$  is grain size and  $\beta$  is FWHM.

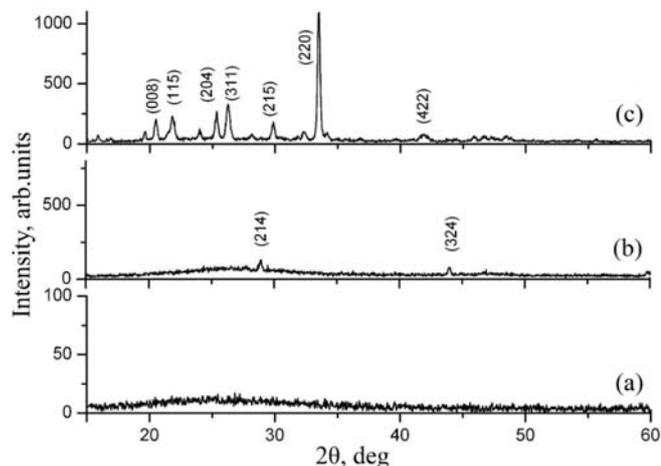


Fig.1. Typical XRD spectra of  $\text{In}_2\text{S}_3$  thin films: before annealing (a), after annealing at  $330^\circ\text{C}$  (b) and at  $400^\circ\text{C}$  (c)

With annealing temperature increasing, the grain size increased from 50 to 75 nm. The smaller grain size for the films annealed at  $330^\circ\text{C}$  can be due low mobility and diffusion of the deposited molecules because of lower thermal energy.

Fig. 2 shows Raman spectra of  $\text{In}_2\text{S}_3$  thin films deposited on glass substrates and annealed at  $330^\circ\text{C}$  and  $400^\circ\text{C}$ . After deposition and after annealing at  $330^\circ\text{C}$ , the Raman peaks at  $164$  and  $281\text{ cm}^{-1}$  as well as low-intensity peaks at  $95$  and  $131\text{ cm}^{-1}$  indicated the presence of  $\beta\text{-In}_2\text{S}_3$  defect spinel, tetragonal, structure [22-23]. Raman peaks at  $126$ ,  $244$  and  $260\text{ cm}^{-1}$  corresponding to  $\alpha\text{-In}_2\text{S}_3$  cubic

structure rise after annealing of  $\text{In}_2\text{S}_3$  films at  $400^\circ\text{C}$  [24, 25]. This result demonstrates that the annealing leads to transformation from  $\beta\text{-In}_2\text{S}_3$  to  $\alpha\text{-In}_2\text{S}_3$  phase at temperatures about  $400^\circ\text{C}$ .

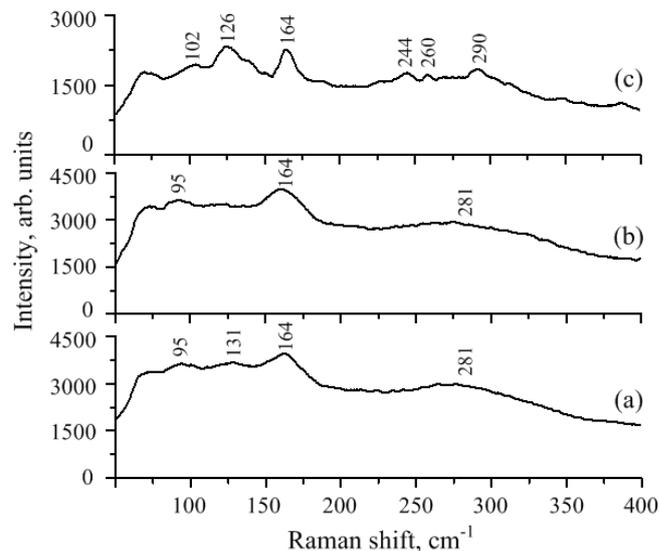


Fig.2. Typical Raman spectra of  $\text{In}_2\text{S}_3$  thin films: before annealing (a), after annealing at  $330^\circ\text{C}$  (b) and at  $400^\circ\text{C}$  (c)

Fig. 3 shows typical 2D and 3D AFM images of  $\text{In}_2\text{S}_3$  thin films before and after annealing at  $330^\circ\text{C}$  and at  $400^\circ\text{C}$ . The  $\text{In}_2\text{S}_3$  thin films are continuous and homogeneous. It is observed from the images that the average height of the roughness profile increases with annealing temperature. The as-deposited films have a granular surface morphology. However, in case of the films annealed at different temperatures, the surface had completely different morphology.

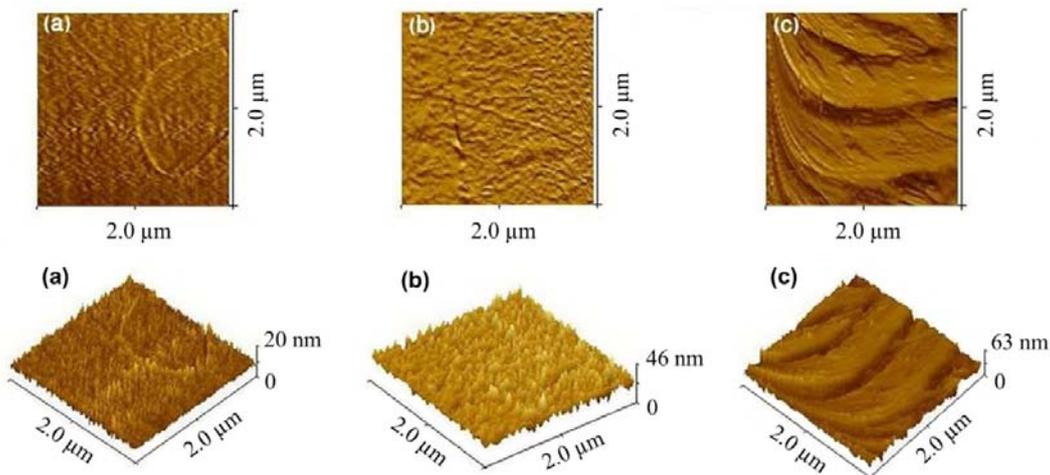


Fig.3. Typical 2D and 3D AFM images of  $\text{In}_2\text{S}_3$  thin films: before annealing (a), after annealing at  $330^\circ\text{C}$  (b) and at  $400^\circ\text{C}$  (c)

Surface roughness analysis of  $\text{In}_2\text{S}_3$  thin films is presented in Table. The results of the statistical calculation revealed that the height distribution has changed after annealing of the samples. It can be seen that the surface roughness increases with the increase of annealing

temperature. Before annealing, the average roughness of thin films was of  $\sim 1.3\text{ nm}$ . But when the samples were annealed at  $400^\circ\text{C}$ , the average roughness increased to  $\sim 4.0\text{ nm}$ .

Table: Average values obtained from AFM measurements and defined roughness of  $\text{In}_2\text{S}_3$  thin films

Parameter	Without annealing	Annealed at	
		330°C	400°C
Arithmetic average height, nm	9.40	24.42	31.17
Average roughness $R_a$ , nm	1.30	3.74	3.97
Root mean square roughness $R_{rms}$ , nm	1.68	4.76	5.06
Skewness $R_{sk}$ , nm	0.35	-0.46	-0.42
Kurtosis $R_{ku}$ , nm	0.91	0.65	0.94

The increase of  $R_a$  and  $R_{rms}$  as the annealing temperature increased corresponds to the improving of  $\text{In}_2\text{S}_3$  films crystalline quality [26-28]. In Table, change of skewness sign from positive to negative indicates that the formation of protruding grains occurred during the annealing. The kurtosis is a yardstick for the sharpness of a surface, and expresses the sharpness of the height distribution; the height distribution is not sharp at  $R_{ku} < 3$  [26]. The results of skewness and kurtosis analysis (Table) showed that the surface of  $\text{In}_2\text{S}_3$  film becomes generally rough during the annealing, with protruding grains being dominant.

### Conclusions

Correlation between the structural properties and the annealing temperature of  $\text{In}_2\text{S}_3$  thin films was reported. The Raman peaks of  $\text{In}_2\text{S}_3$  thin films indicating the presence of  $\beta$ - $\text{In}_2\text{S}_3$  in the as-deposited samples and after annealing at 330°C. After annealing at 400°C, new peaks related to  $\alpha$ - $\text{In}_2\text{S}_3$  were observed. Phase transitions after annealing at 400°C resulting in co-existence of both cubic and tetragonal phases in  $\text{In}_2\text{S}_3$  layers was also detected by XRD analysis. It was found that the increase of annealing temperature leads to greater crystallite sizes, as well as the increased roughness of the films. The values of surface roughness and grain size can be controlled by adjusting the annealing temperature.

These results provide a more comprehensive understanding of the annealing temperature influence on  $\text{In}_2\text{S}_3$  films structural and morphological features and could help in controlling the topographical parameters according to the surface morphology requirements suitable for photovoltaic applications.

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